# DEVELOPMENT OF MULTIFUNCTIONAL ADDITIVE COMBINED ELECTROSPUN CARBON NANO FIBERS INTEGRATED BIMETALLIC COPPER COBALT PHOSPHIDE AS AN INTERFACIAL LAYER FOR HIGH-PERFORMANCE DSSC

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#### ABSTRACT

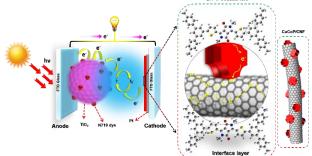
This study developed a star-shaped 3,6-Bis(5-(4,4' - bis(3-azidopropyl)-[1,1' :3' ,1' -terphenyl]-5' -yl)-thien-2-yl)-2,5-bis(2-ethylhexyl)pyrrolo[3,4-c]pyrrole-

1,4(2H,5H)-dione (DPPTPTA) additive in combination with electrospun carbon nano fibers integrated bimetallic copper-cobalt phosphide (DPPTPTA/CuCoP/CNF) as an interface layer for dyesensitized solar cells (DSSCs). This interface layer is different from conventional blend films. As a multifunctional additive, DPPTPTA provides ladder-like energy levels for efficient charge separation, changing the morphology of the interface layer and improving performance. The CuCoP/CNF has high electrocatalytic activity and enhanced ionic conductivity. The interface layer ensures efficient charge transfer at the electrode/electrolyte interface with lower electron recombination rate in the DSSC, and it is expected that the photoelectric conversion efficiency  $(\eta)$  of the DSSC can be greatly improved under the condition of 1 sun (AM 1.5G, 100 mW cm<sup>-2</sup>), and the local boundary is constructed through the interface layer. It ensures longterm thermal stability; and it is expected to have higher  $\eta$  under indoor light condition test, which is one of the most promising devices for renewable energy generation.

**Keywords:** Bimetallic phosphides, conjugated small polymer, dye-sensitized solar cells, electrospun, interface layer, renewable energy

#### NONMENCLATURE

Abbreviations				
CNF	Carbon nano fibers			
CuCoP	Copper-cobalt phosphide			
DSSCs	Dye-sensitized solar cells			
PtCE	Platinum counter electrode			
Symbols				
ή	Photoelectric conversion efficiency			
Voc	Open-circuit voltage			
J <sub>SC</sub>	Short-circuit current density			
FF	Fill factor			



Scheme 1 The working mechanism of **DPPTPTA/CuCoP/CNF** acts as an interface layer in DSSCs.

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# 1. INTRODUCTION

The solar cells provide an effective way to convert the solar energy into electricity directly. Among various kinds of solar cells, the DSSC is especially important in contrast with the conventional solar cells due to its low manufacturing cost [1]. DSSC is comprised mainly of titanium dioxide photoanode (TiO<sub>2</sub>) with adsorbed dye molecules, electrolyte containing redox species, and the electrocatalytic platinum counter electrode (PtCE).

Traditionally, liquid electrolyte consists of redox couple  $(I^{-}/I_{3}^{-})$  in organic solvent yields  $\eta$  up to 12%. However, the technological issues due its undesirable intrinsic properties such as solvent evaporation, dye degradation, and leakage affect its long-term stability, which are restrict the commercialization of DSSCs. To solve these problems, several attempts have been made to replace liquid electrolyte with p-type semiconductors, molten salts, organic hole transport materials, and ionic conducting polymers [2]. Nevertheless, the solid state electrolytes show the low electron injection efficiency and the imperfection of filling materials into the nanosized pores of the anode to form a close contact at the interface. Moreover, the absence of solvent leads to deterioration of cell due to the crystallization of salt [3]. To encounter the problems of both liquid and solid electrolytes, the gel polymer electrolytes have been focused intensively due to their stability. However, the gel electrolyte has a low electrolyte load per unit volume and therefore has a low ionic conductivity. Due to the poor interfacial contact, the PtCE/gel polymer interface has a higher resistance than the CE/liquid electrolyte interface for the electrocatalytic reaction of triiodide with reflowed electrons [4]. This leads to a reduction in the overall n. In view of these disadvantages, electrospun polymer membranes activated in ionic liquid electrolytes have been studied for DSSC applications [5].

Here, we propose a multifunctional interface layer that combines the conjugated small molecule **DPPTPTA** and the **CuCoP/CNF** to resolve the problems encountered by the traditional DSSCs mentioned above. A new design for multifunctional additives, namely, **DPPTPTA**, which is easy to synthesize from a diketopyrrolopyrrole (DPP) core. And also, many features to easy synthesis including highly reactive endcapping azido groups and thermal behavior suitable for triggering cross-linking reactions. **DPPTPTA** has a broad absorption wavelength, a favorable absorption intensity, high charge mobility, and a terphenylbased conjugated  $\pi$ -bridge, which can efficiently inhibit molecular aggregation, and induce optimal morphology for the multicomponent system [7]. After incorporating **DPPTPTA** into the interface layer, an enhancement in device performance and thermal stability can be observed [8]. First, some metal compounds exhibit high conductivity (e.g., copper compounds) and some display great electrochemical activity (e.g., cobalt compounds) [9]. Second, the type of anion in metal compounds is another effective agent. By selecting anions with lower electronegativity than oxygen, the electrical conductivity and electrochemical activity of electrodes can be significantly increased [10]. Recently, strong interest has been devoted to the development of transition metal phosphide electrodes for advanced energy storage and conversion applications [11]. Third, an effective approach to enhance the surface area and conductivity is structural engineering. The morphologies that possess nanosized walls with numerous nanopores can provide a high surface area and reduced length for both mass and charge transport [12].

# 2. EXPERIMENTAL SECTION

### 2.1 Materials

Lithium perchlorate (LiClO<sub>4</sub>,  $\geq$ 98.0%), titanium(IV) tetraisoproproxide (TTIP, >98%), 2-methoxyethanol 3,6-bis(5-bromo-2-thienyl)-2,5-bis(2-(99.95%). hexyldecyl)-2,5-dihydro-pyrrolo[3,4-c]pyrrole-1,4-dione (C<sub>46</sub>H<sub>70</sub>Br<sub>2</sub>N<sub>2</sub>O<sub>2</sub>S<sub>2</sub>, 98%), 4-(4,4,5,5-tetramethyl-1,3,2dioxaborolan-2-yl)toluene  $(C_{13}H_{19}BO_2),$ 2-azido-1,3dimethylimidazolinium hexafluorophosphate (C<sub>5</sub>H<sub>10</sub>F<sub>6</sub>N<sub>5</sub>P, 97%), 1,8-diazabicyclo[5.4.0]undec-7-ene  $(C_9H_{16}N_2, 98\%)$ , sodium carbonate  $(Na_2CO_3, \ge 99.0\%)$ , diethyl ether (≥99.9%). The 1,4-dioxane (anhydrous, 99.8%), toluene (anhydrous, 99.8%), copper(II) chloride dehydrate (CuCl<sub>2</sub> · 2H<sub>2</sub>O,  $\geq$ 99.0%), cobalt(II) chloride hexahydrate (CoCl<sub>2</sub> · 6H<sub>2</sub>O, 98%) and red phosphorous (≥99.99%) were obtained from Sigma-Aldrich. Lithium iodide (Lil, synthetical grade), poly(ethylene glycol) (PEG, MW~20,000) and iodine (I<sub>2</sub>, synthetical grade) were received from Merck. 4-tert-butylpyridine (tBP, 96%) and tert-butyl alcohol (tBA, 96%) were procured from Acros. 3-Methoxypropionitrile (MPN, 99%) was bought from Fluka.

# 2.2 Synthesis of the **DPPTPTA** conjugated small molecule

The precursor of **DPPTPTA** was synthesized through Suzuki–Miyaura cross-coupling of 3,6-bis(5-bromothien-2-yl)-2,5-bis(2-ethylhexyl)pyrrolo[3,4-c]-pyrrole-

1,4(2H,5H)-dione and 3,3 ' -(5 ' -(4,4,5,5-tetramethyl-

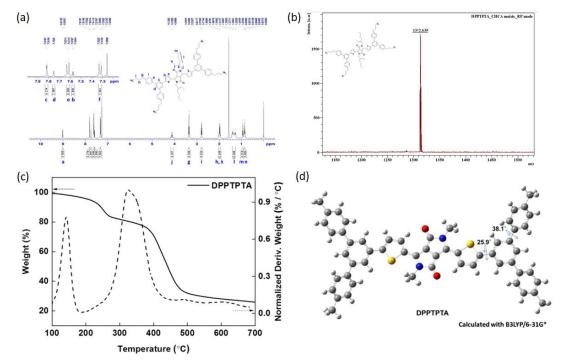


Fig. 1. (a) Chemical positioning determined from FT-NMR spectrum, (b) MALDI-TOF mass spectrum, (c) thermal properties, and (d) optimized structures, calculated with B3LYP/6-31G\* theoretical computational models, of **DPPTPTA**.

1,3,2-dioxaborolan-2-yl)-[1,1 ' :3 ' ,1 ' -terphenyl]-4,4 ' -diyl)bis-(propan-1-ol) [13]. **DPPTPTA** was obtained through anazide exchange, reacting with 2-azido-1,3dimethylimidazolinium hexafluorophosphate (ADMP) and 1,8-diazabicyclo-[5.4.0]undec-7-ene (DBU) [14].

# 2.3 Synthesis of (DPPTPTA/CuCoP/CNF)

The **DPPTPTA/CuCoP/CNF** was prepared through a single-nozzle electrospinning technique followed by controlled high temperature treatment. In the typical preparation, 1 g of PAN was dissolved in 25 mL of DMF under 80°C. Then 0.75 g of copper(II) chloride dehydrate, cobalt(II) chloride hexahydrate, and red phosphorous were dispersed in 25 mL of DMF for 2 h under magnetic stirring. The dispersed metal precursors were gently added to the homogenous mixture of PAN/DMF solution and conserved with vigorous agitation for 6 h under 80°C to obtain a homogeneous polymer composite precursor. Subsequently, the precursor solution was loaded into a 15 mL plastic syringe equipped with a 21 - gauge blunt tip needle. The flow rate of solution was set to 10  $\mu$ L min<sup>-1</sup> by a syringe pump.

The needle was connected to a high - voltage power supply to apply a voltage of 15 kV between the needle and the collector for the formation of nanofibers. The distance maintained between the needle and the collector was 15 cm. Finally, the as-collected film was initially pacified at 80°C for 6 h in an oven and further stabilized at 250°C for 2 h with a heating rate of 10 °C min<sup>-1</sup>. Followed by carbonization at 900°C for 3 h in N<sub>2</sub> atmosphere with a heating rate of 10 °C min<sup>-1</sup>, and combined with different concentrations of **DPPTPTA** to prepare the interface layer of DSSCs on PtCE by drop coating.

#### 2.4 Cell assembly

A cleaned FTO glass was coated with a thin compact layer of TiO<sub>2</sub> (100 nm), by using a solution of TTIP in 2methoxyethanol (weight ratio = 1/3). A TiO<sub>2</sub> film, containing a first layer of transparent layer (10  $\mu$ m) and a second layer of scattering layer (4  $\mu$ m) with an active area of 0.20 cm<sup>2</sup> was coated on the treated FTO glass by using a doctor blade technique. The transparent layer was coated on the compact layer using the abovementioned commercial transparent paste (Ti-nanoxide T/SP), while the scattering layer was further coated on the transparent layer using a home-made scattering paste. Each layer was separately sintered at 500 °C for 30 min in ambient environment. After the sintering process, the TiO<sub>2</sub> film was immersed in a 5×10<sup>-4</sup> M N719 dye

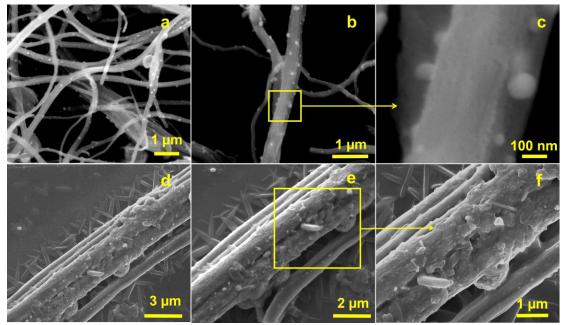


Fig. 2. Different magnified FE-SEM images of (a-c) CuCoP/CNF and (d-f) DPPTPTA/CuCoP/CNF.

Table 1. Photovoltaic properties of different type of devices in DSSCs.							
Type of devices	η (%)	<i>Voc</i> (mV)	J <sub>sc</sub> (mA cm <sup>-2</sup> )	FF	V <sub>onset</sub> (mV)		
Standard device	7.88 ± 0.08	761 ± 2	15.62 ± 0.06	0.66 ± 0.01	520		
Bare DPPTPTA	8.53 ± 0.02	793 ± 1	15.86 ± 0.03	0.68 ± 0.00	595		
Bare CuCoP/CNF	8.59 ± 0.03	813 ± 1	15.61 ± 0.02	$0.68~\pm~0.01$	608		
DPPTPTA/CuCoP/CNF	9.50 ± 0.02	827 ± 2	$16.25 \pm 0.02$	0.71 ± 0.00	655		

Table 1. Photovoltaic properties of different type of devices in DSSCs

✗ The standard device means without an interface layer. ▮

times The standard deviation data for each DSSC are obtained based on three cells.

solution for 24 h at room temperature using a mixed solvent of tBA and ACN (volume ratio = 1/1).

Thus, a dye-adsorbed TiO<sub>2</sub> film was prepared as the photoanode of a DSSC. Finally, the dye-adsorbed TiO<sub>2</sub> photoanode was coupled with a PtCE with an interface layer, and a cell gap of 25  $\mu$ m thick Surlyn<sup>®</sup> film as the spacer. The electrolytes in this study were composed of 0.1 M Lil, 0.05 M I<sub>2</sub>, 0.6 M DMPII, 0.5 M *t*BP in ACN/MPN (volume ratio = 8/2). These electrolytes were injected into the cell gap between the two electrodes by capillarity. Finally, a DSSC of a photoanode/electrolyte/interface layer/PtCE is formed.

# 3. RESULTS AND DISCUSSION

# 3.1 Characterization of **DPPTPTA** conjugated small molecule

The compounds were characterized using elemental analysis, FT-NMR spectroscopy, and matrix-assisted laser desorption/ionization time-of-flight mass spectrometry (MALDI-TOF MS). We assigned the chemical shifts for each proton in the FT-NMR spectra (Fig. 1a) and confirmed the high purity of **DPPTPTA** from its MALDI-TOF mass spectrum (Fig. 1b). Thermogravimetric analysis (TGA) revealed that DPPTPTA exhibited two stages of thermal decomposition at 142 and 325 °C; the first involves thermal decomposition of the azido groups and the second characterizes the thermal degradation of the conjugated backbone (Fig. 1c). Theoretical calculations suggested that DPPTPTA should possess the large bondtwisting angles (25.9 and 38.1°), and the terphenyl group should induce strong steric effects (Fig. 1d) leading to a low degree of intermolecular stacking. As we had expected, the incorporation of the terphenyl structure led to the molecular twisting of the DPP derivative. The efficient intermolecular  $\pi$ -stacking of DPP was undermined, so that the resultant **DPPTPTA** exhibits a lower degree of molecular aggregation.

### 3.2 Characterization of CuCoP/CNF

Fig. 2(a-c) shows the field-emission scanning electron microscope (FE-SEM) images of **CuCoP/CNF**. It suggests that the **CuCoP/CNF** has the uniform threedimensional interconnected carbon network, whereas the uniform structure of nanoparticle can be observed on the CNF structure. In addition, the integration of **DPPTPTA** with **CuCoP/CNF** is evidently confirmed by using the FE-SEM images (Fig. (d-f)). It shows that the **DPPTPTA** is uniformly covered on the surface of the **CuCoP/CNF** and acts as an interconnected path between the fibers. It facilitates the faster charge carrier conduction to redox couples from the PtCE.

Fig. 3a displays the XRD pattern of CuCoP/CNF, copper phosphide, cobalt phosphide, and graphite samples. The obtained XRD pattern of copper phosphide shows the peaks to the standard JCPDS 65-1973. Whereas, the cobalt phosphide exhibited the diffraction peaks, which are matched with JCPDS 65-2381. At the same time, the XRD pattern of graphitic carbon shows only the single peak at 13.09° for corresponding plane of (210) related to the JCPDS 8-415 standard data. Finally, **CuCoP/CNF** composite exhibits the combined diffraction peaks of copper phosphide, cobalt phosphide and graphitic carbon. It preliminarily suggests the formation of CuCoP and its integration of graphitic CNF. the XPS study was carried out for the prepared CuCoP/CNF and recorded the resultant wide scan and high-resolution spectra for Cu 2p, Co 2p, P 2p and C 1s (Fig. 3b). The wide scan XPS spectra is shown in Fig. 3(c-f) with the characteristic peaks for Cu 2p, Co 2p, P 2p, C 1s and O 1s. It concluded the presence of Cu 2p, Co 2p, P 2p, and C 1s states in CuCoP/CNF samples. Meanwhile, the O 1s is ascribed due to the surface contaminants. The Fig. 3c shows the high-resolution XPS spectra of Cu 2p, which exhibits the two peaks for spin-orbital doublets of Cu  $2p_{1/2}$  and Cu  $2p_{3/2}$  at binding energies of 952.46 and 932.65 eV respectively. Both the doublet peaks contain the mixed electronic states of +2 and +4. Moreover, the two satellite peaks were detected at 942.44 and 962.37 eV. Meanwhile, the XPS spectra of Co 2p deconvoluted into two characteristics peaks of Co  $2p_{1/2}$  (796.93 eV) and Co  $2p_{3/2}$  (780.84 eV) including the two satellite peaks at binding energies of 786.05 and 802.57 eV respectively (Fig. 3d). The high resolution XPS spectra of P 2p shows

in Fig. 3e, from the spectra the peak at 129.377 eV is associated to P  $2p_{1/2}$  and P  $2p_{3/2}$ . Whereas the peak at 132.97 eV is associated to the P due to the exposure of atmosphere air. Besides, the characteristic peaks located at 287.54, 285.55 and 284.95 eV can be ascribed due to the C=C-O, C-O and C-C species respectively (Fig. 3f).

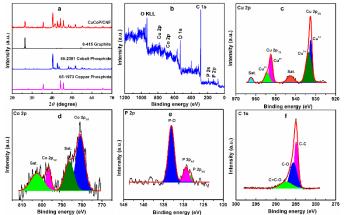


Fig. 3. (a) XRD pattern (b) wide scan XPS (c-f) wide scan XPS spectra of **CuCoP/CNF**.

# 3.3 Photovoltaic performance of DSSCs with interface layer

The photocurrent density-voltage (J-V) curves for a standard DSSC with and without the interface layers obtained under the condition of 1 sun (AM 1.5G, 100 mW cm<sup>-2</sup>) are shown in Fig. 4, and their corresponding values are given in Table 1. When iodide and polyiodide ions adsorbed on the surface of bimetallic CuCoP nanoparticles, the electronic conductivity and surface activity of the interface layer enhanced the charge exchange rate. The interface layer possesses both the advantages of a gel polymer electrolyte as well as the diffusivity of the liquid electrolytes, since its interconnected pore structure can easily entrap an enormous quantity of liquid electrolyte. Thus, the stability of the DSSC with the interface layer is far superior to the conventional DSSC.

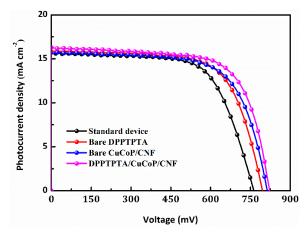


Fig. 4. Current density-voltage curves of the DSSCs with different contents of interface layers measured at 1 sun illumination.

# 4. CONCLUSIONS

The DPPTPTA/CuCoP/CNF is the first application as an interface layer in DSSCs with high ionic conductivity, thermal stability, and good electrochemical properties. The strong interaction between the DPPTPTA/CuCoP/CNF and liquid electrolyte induces the space charge layer, which facilitates the fast electron conduction and enhancing the  $I^{-}/I_{3}^{-}$  shuttle. In this study, the DSSC combined with DPPTPTA/CuCoP/CNF interface layer exhibited the good photovoltaic performance and achieved the highest  $\eta$  of 9.50%. The obtained  $\eta$  is comparatively higher than the standard device without the interface layer (7.88%). Finally, a new type of interface layer was developed in this study. This approach can be used not only in the field dye-sensitized solar cells, but also for the organic solar cells.

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